

# Ultrahigh field electron cyclotron resonance absorption in $\text{In}_{1-x}\text{Mn}_x\text{As}$ films

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(Dated: February 1, 2008)

We have carried out an ultrahigh field cyclotron resonance study of  $n$ -type  $\text{In}_{1-x}\text{Mn}_x\text{As}$  films, with Mn composition  $x$  ranging from 0 to 12%, grown on GaAs by low temperature molecular beam epitaxy. We observe that the electron cyclotron resonance peak shifts to lower field with increasing  $x$ . A detailed comparison of experimental results with calculations based on a modified Pidgeon-Brown model allows us to estimate the  $s$ - $d$  and  $p$ - $d$  exchange coupling constants,  $\alpha$  and  $\beta$ , for this important III-V dilute magnetic semiconductor system.

$\text{InMnAs}$  alloys and their heterostructures with  $\text{AlGaSb}$ , the first grown III-V dilute magnetic semiconductor (DMS) [1, 2, 3], serve as a prototype for implementing electron and hole spin degrees of freedom in semiconductors. Recent experiments have demonstrated the feasibility of controlling ferromagnetism in these systems optically [4] and electrically [5]. Understanding their electronic, transport, and optical properties is crucial for designing novel ferromagnetic semiconductor devices with high Curie temperatures. However, their basic band parameters such as effective masses and  $g$ -factors have not been accurately determined.

The localized Mn spins strongly influence the delocalized conduction and valence band states through the  $s$ - $d$  and  $p$ - $d$  exchange interactions. These interactions are usually parameterized as  $\alpha$  and  $\beta$ , respectively [6]. Determining these parameters is important for understanding the nature of Mn electron states and their mixing with delocalized carrier states. In narrow gap semiconductors like  $\text{InMnAs}$ , due to strong interband mixing,  $\alpha$  and  $\beta$  can influence *both* the conduction and valence bands. This is in contrast to wide gap semiconductors where  $\alpha$  influences primarily the conduction band and  $\beta$  the valence band. In addition, in narrow gap semiconductors, due to the strong interband mixing, the coupling to the Mn spins does not effect all Landau levels by the same amount. As a result, the electron cyclotron resonance (CR) peak can

shift as a function of  $x$ , the Mn concentration. This can be a sensitive method for estimating these exchange parameters. A recent CR study on  $\text{Cd}_{1-x}\text{Mn}_x\text{Te}$  [7] showed that the electron mass is strongly affected by  $sp-d$  hybridization. To our knowledge, however, there have been no CR studies on any III-V DMS systems (for our preliminary results, see [8, 9]).

In this paper we describe the dependence of electron CR observed in  $n$ -type  $\text{In}_{1-x}\text{Mn}_x\text{As}$  films on  $x$ . A detailed comparison with theoretical calculations provides insight into the effects of Mn ions on the Landau and Zeeman splittings of the conduction band states.

TABLE I: Densities, mobilities, and cyclotron masses for the four samples studied. The densities are in units of  $\text{cm}^{-3}$  and the mobilities  $\text{cm}^2/\text{Vs}$ . The masses were obtained at a photon energy of 117 meV (or  $\lambda = 10.6 \mu\text{m}$ ).

Mn content $x$	0	0.025	0.050	0.120
Density (4.2 K)	$\sim 1.0 \times 10^{17}$	$1.0 \times 10^{16}$	$0.9 \times 10^{16}$	$1.0 \times 10^{16}$
Density (290 K)	$\sim 1.0 \times 10^{17}$	$2.1 \times 10^{17}$	$1.8 \times 10^{17}$	$7.0 \times 10^{16}$
Mobility (4.2 K)	$\sim 4000$	1300	1200	450
Mobility (290 K)	$\sim 4000$	400	375	450
$m/m_0$ (30 K)	0.0342	0.0303	0.0274	0.0263
$m/m_0$ (290 K)	0.0341	0.0334	0.0325	0.0272

We studied four  $\sim 2\text{-}\mu\text{m}$ -thick  $\text{In}_{1-x}\text{Mn}_x\text{As}$  films with  $x = 0, 0.025, 0.050$  and  $0.120$  by ultrahigh-field magneto-absorption spectroscopy. The films were grown by molecular beam epitaxy on semi-insulating GaAs substrates at  $200^\circ\text{C}$ . All the samples were  $n$ -type and did not show ferromagnetism down to 1.5 K. The electron densities and mobilities deduced from Hall measurements are listed in Table I. The single-turn coil technique [10] was used to generate ultrahigh magnetic fields with pulse duration

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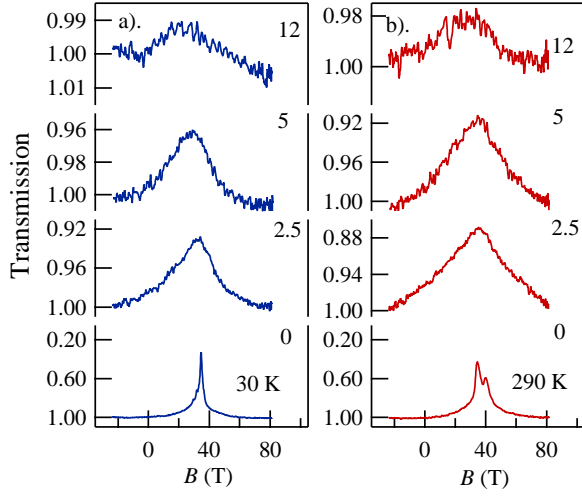


FIG. 1: Experimental CR spectra for different Mn contents taken for 290 (a) and 30 K (b). The wavelength of the laser is fixed at  $10.6 \mu\text{m}$  with electron-active circular polarization and the field  $B$  is swept. The resonance position shifts to lower  $B$  with increasing  $x$ . The  $x$  values are 0%, 2.5%, 5% and 12%.

$\sim 7 \mu\text{s}$ . The sample and pick-up coil were placed in a helium flow cryostat. We used the  $10.6 \mu\text{m}$  line from a  $\text{CO}_2$  laser and produced circular polarization using a CdS quarter-wave plate. The transmitted radiation was detected by a fast HgCdTe photovoltaic detector.

Typical measured CR spectra at 30 K and 290 K are shown in Figs. 1 (a) and (b), respectively. Note that to compare the transmission with absorption calculations, the transmission increases in the negative  $y$  direction. Each figure shows spectra for all four samples labeled by the corresponding Mn compositions from 0 to 12 %. All the samples show pronounced absorption peaks (or transmission dips) and the resonance field *decreases* with increasing  $x$ . Increasing  $x$  from 0 to 12% results in a  $\sim 25$  % decrease in cyclotron mass (see Table I). It is important to note that at resonance, the densities and fields are such that only the lowest Landau level for each spin type is occupied (see Fig. 2). Thus, all the electrons were in the lowest Landau level for a given spin even at room temperature, precluding any density-dependent mass due to nonparabolicity (expected at zero or low magnetic fields) as the cause of the observed trend.

At high temperatures (e.g., Fig. 1(b)) the  $x = 0$  sample clearly shows nonparabolicity-induced CR spin-splitting with the weaker (stronger) peak originating from the lowest spin-down (spin-up) Landau level, while the other three samples do not show such splitting. The reason for the absence of splitting in the Mn-doped samples is a combination of 1) their low mobilities (which lead to substantial broadening) and 2) the large effective  $g$ -factors due to the Mn ions; especially in samples with large  $x$  only the spin-down level is substantially thermally populated (cf. Fig. 2).

We also performed midinfrared interband absorption measurements at various temperatures using Fourier-

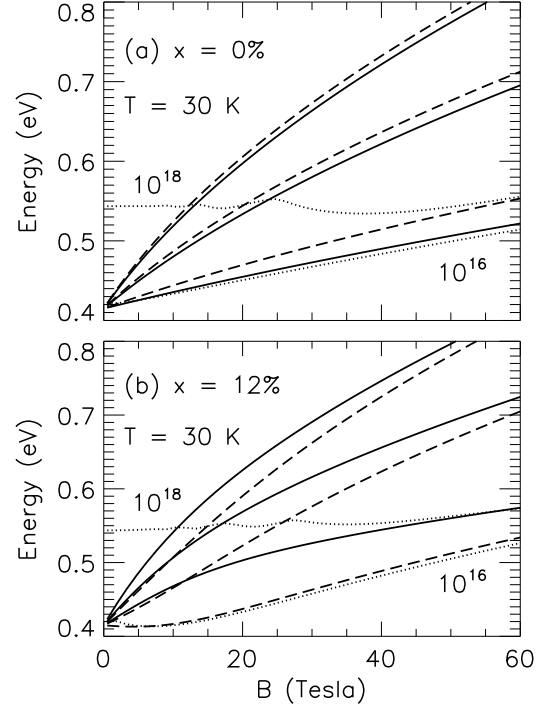


FIG. 2: Electron Landau levels as a function of magnetic field,  $B$ , at 30 K for (a)  $x = 0\%$  and (b)  $x = 12\%$ . Spin up (down) levels are shown as solid (dashed) lines. The Fermi levels are also indicated for electron concentrations of  $10^{16}$  and  $10^{18} \text{ cm}^{-3}$ .

transform infrared spectroscopy, which revealed no significant  $x$ -dependence of the band gap except for the different amounts of Burstein-Moss shifts due to the different carrier densities.

We calculated the conduction band Landau levels based on a modified Pidgeon-Brown model [11, 12] including full  $k_z$  dependence and also  $sp$ - $d$  exchange coupling of the electrons and holes to the Mn ions. This is an  $8 \times 8$  band  $\mathbf{k} \cdot \mathbf{p}$  method with a magnetic field along the [001] direction. We used a standard set of band parameters for InAs [13], neglecting any  $x$ -dependence of these parameters. We vary the gap with  $T$  according to  $E_g(T) = 0.417 - 0.000276T^2/(93 + T)\text{eV}$  [13]. As seen in Table I, the mass does not vary significantly with  $T$ . Other low field CR studies of InAs [14] show a very weak  $T$ -dependence to the cyclotron mass and also take into account polaron effects. Since we are mainly interested in the  $x$  dependence of the mass, for simplicity, we adjust the  $F$  parameter to keep the mass constant with  $T$ . Alternatively, one might wish to keep  $F$  constant and vary the  $\mathbf{k} \cdot \mathbf{p}$  optical matrix element  $E_p$ .

Since InMnAs is a narrow gap semiconductor, the coupling between the valence and conduction bands is strong, and thus, not only  $\alpha$  but also  $\beta$  is important in calculating the conduction band Landau levels. A pho-

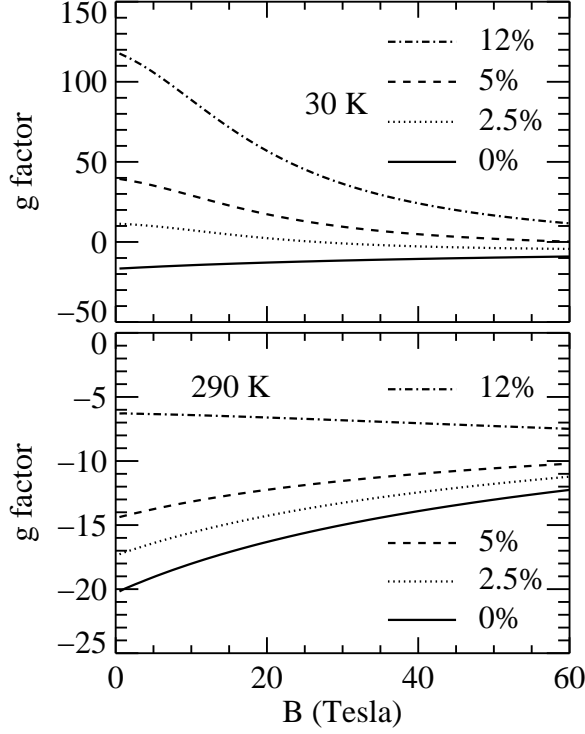


FIG. 3: Electron effective  $g$  factors for  $In_{1-x}Mn_xAs$  as a function of applied magnetic field for several values of the  $Mn$  concentration,  $x$ . The upper and lower panels correspond to  $T = 30$  K and  $T = 290$  K, respectively.

to emission experiment [15] reports  $\beta = -0.7$  eV, and a theoretical estimate [16] suggests that  $\beta \approx -0.98$  eV. No value has been reported for  $\alpha$ , to our knowledge.

We chose  $\beta = -1.0$  eV and  $\alpha = 0.5$  eV as the values which best represent the observed trends though these values can change depending upon the other parameters used in the  $\mathbf{k} \cdot \mathbf{p}$  calculation. Figures 2(a) and 2(b) show the lowest three Landau levels (spin up and down) in the conduction band for the  $x = 0$  and 12% samples. In addition, the Fermi levels are calculated and plotted (dotted lines) for carrier densities of  $10^{18} \text{ cm}^{-3}$  and  $10^{16} \text{ cm}^{-3}$ . Both the mass and  $g$ -factor are strongly energy and magnetic field dependent as well as  $x$  dependent. We can also see that in the  $x = 12\%$  sample the *spin order is reversed* and the spin splitting is significantly enhanced by the presence of Mn ions; the latter explains the absence of CR spin splitting in the 12% sample. We found that both the sign and values of  $\alpha$  and  $\beta$  are critically important in explaining the experimental data and therefore it is a good method for determining these parameters. In various investigations of II-VI DMS's it has been established that  $\alpha$  and  $\beta$  differ in sign and usually the absolute value of  $\beta$  is greater than  $\alpha$  [6, 12], which is consistent

with our results.

In Fig. 3, we show the effective  $g$ -factors for electrons in the lowest Landau level in  $In_{1-x}Mn_xAs$  as a function of applied magnetic field and Mn doping concentration,  $x$ . The curves are shown for temperatures of 30 K and 290 K, respectively. As can be seen, at 30 K and high Mn concentration, the effective  $g$ -factor is extremely large ( $> 100$ ) and has the opposite sign of the undoped sample. Due to Mn doping and the large nonparabolicity of the narrow gap semiconductor, the  $g$ -factor is very dependent on magnetic field and is reduced substantially at high magnetic fields and temperatures.

Based on the calculated Landau levels, we also calculated the cyclotron resonance spectra for the four samples at 30 K and 290 K. The magneto-optical absorption coefficient at the photon energy  $\hbar\omega$  is [17]

$$\alpha(\hbar\omega) = \frac{\hbar\omega}{(\hbar c)n_r} \epsilon_2(\hbar\omega) \quad (1)$$

where  $\epsilon_2(\hbar\omega)$  is the imaginary part of the dielectric function and  $n_r$  is the index of refraction. The imaginary part of the dielectric function is found using Fermi's golden rule. The result is

$$\epsilon_2(\hbar\omega) = \frac{e^2}{\lambda^2(\hbar\omega)^2} \sum_{n,\nu;n',\nu'} \int_{-\infty}^{\infty} dk_z |\hat{e} \cdot \vec{P}_{n,\nu}^{n',\nu'}(k_z)|^2 \times (f_{n,\nu}(k_z) - f_{n',\nu'}(k_z)) \delta(\Delta E_{n,\nu}^{n',\nu'}(k_z) - \hbar\omega), \quad (2)$$

where  $\Delta E_{n,\nu}^{n',\nu'}(k_z) = E_{n',\nu'}(k_z) - E_{n,\nu}(k_z)$  is the transition energy and  $\lambda$  the magnetic length. The function  $f_{n,\nu}(k_z)$  in Eq. (2) is the probability that the state  $(n, \nu, k_z)$ , with energy  $E_{n,\nu}(k_z)$ , is occupied. It is given by the Fermi distribution function and depends on the net carrier density.

Figures 4(a) and 4(b) show the calculated CR absorption coefficient for electron-active circularly polarized 10.6  $\mu\text{m}$  light in the Faraday configuration as a function of magnetic field at 30 K and 290 K, respectively. Densities for each sample are given in Table I. In the calculation, the curves were broadened based on the mobilities of the samples. The broadening used was: ( $T = 30$  K, 0% - 4 meV, 2.5% - 40 meV, 5% - 40 meV and 12% - 80 meV); ( $T = 290$  K, 0% - 4 meV, 2.5% - 80 meV, 5% - 80 meV, 12% - 80 meV).

At  $T = 30$  K, we see a shift in the CR peak as a function of doping in agreement with Fig. 1 (a). For  $T = 290$  K, we see the presence of two peaks in the pure InAs sample. The second peak originates from the thermal population of the lowest spin-down Landau level (cf. Fig. 2). The peak does not shift as much with doping as it did at low temperature. This results from the temperature dependence of the average Mn spin. We believe that the Brillouin function used for calculating the average Mn spin becomes inadequate at large  $x$  and/or high temperature due to its neglect of Mn-Mn interactions such as pairing and clustering.

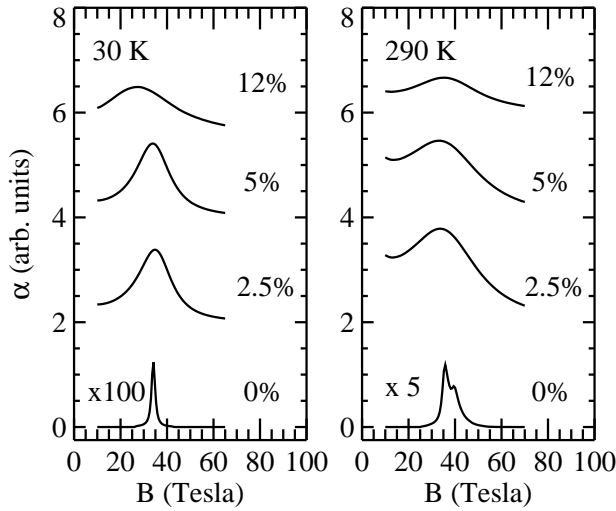


FIG. 4: Calculated CR absorption as a function of magnetic field at 30 K (a) and 290 K (b). The curves are calculated based on the Pidgeon-Brown model and the golden rule for absorption. The curves are broadened based on the mobilities reported in Table I.

In summary, we have observed electron CR in  $\text{In}_{1-x}\text{Mn}_x\text{As}$  for the first time and studied its dependence on the Mn concentration,  $x$ . Owing to band mixing in this narrow gap system, the CR peak actually shifts with  $x$ ; we observed a  $\sim 25\%$  mass reduction by increasing  $x$  from 0 to 0.12. Theoretical calculations based on an  $8 \times 8$  band  $\mathbf{k} \cdot \mathbf{p}$  model successfully reproduce this behavior. These results demonstrate that CR can be used for determining the  $sp-d$ , electron/hole -  $Mn$  ion exchange interactions  $\alpha$  and  $\beta$ . We obtained  $\beta = -1.0$  eV and  $\alpha = 0.5$  eV as the values which best represent the observed trends. We also showed that at low temperatures and high  $x$  the effective  $g$ -factor is extremely large ( $> 100$ ) and has the opposite sign of the  $x = 0$  sample; it is very dependent on magnetic field and reduced substantially at high magnetic fields and temperatures. These findings should be useful for designing novel spin-based semiconductor devices.

This work was supported by DARPA through grant No. MDA972-00-1-0034 and the NEDO International Joint Research Program.

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